

Experimental report

08/09/2015

Proposal: 5-24-542

Council: 4/2014

Title: Evolution of the crystal structure of novel SOFC cathode materials $\text{SrCo}_{1-x}\text{M}_x\text{O}_3$ (M= V,Nb,Ti): unveiling the oxygen diffusion pathway

Research area: Materials

This proposal is a new proposal

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Samples: $\text{SrCo}_{0.95}\text{M}_{0.05}\text{O}_3$ (M= V,Nb,Ti)

Instrument	Requested days	Allocated days	From	To
D2B	3	3	10/10/2014	13/10/2014

Abstract:

The novel perovskites $\text{Sr}_{1-x}\text{Co}_x\text{O}_{3-\delta}$ (M= V,Nb,Ti) have been evaluated as superior cathodes for solid-oxide fuel cells (SOFC); tests in single fuel cells yield output powers higher than 800 mW/cm² at 850 C with H₂ as a fuel. In this experiment we aim to unravel the oxide-ion diffusion path of this family of MIEC (mixed ionic-electronic conductor) oxides. We plan to study the thermal evolution of the crystal structure in order to get information about the actual crystal symmetry, the thermal vibrations and oxygen contents as well as the order-disorder of the oxygen vacancies and their evolution at the actual working conditions in a SOFC cell, in air at temperatures between 600 and 900 C.

The stabilization of a 3D perovskite-like framework in the SrCoO_{3-δ} system has been a widely used strategy in order to obtain an adequate mixed ionic-electronic conductor to be used as cathode in intermediate temperature solid oxide fuel cells. For this purpose, several chemical substitutions have been performed in either the Sr (Ba, La, Sm) [1] or in the Co (Sc, Fe, Ni, etc) [2,3] positions or in both. In this work we have stabilized a perovskite phase by doping the SrCoO_{3-δ} system with low M = V, Nb and Ti contents in SrCo_{1-x}M_xO_{3-δ} (x = 0.05). The stabilization of a tetragonal *P4/mmm* structure was obtained in all cases. The doped compounds present high thermal stability without abrupt changes in the expansion coefficient and a great enhancement of the electrical conductivity compared to the pristine SrCoO_{3-δ} at low and intermediate temperatures (T ≤ 800 °C). The sample SrCo_{0.95}Ti_{0.05}O_{3-δ} displayed the highest conductivity value over 80 S·cm⁻¹[4].

In this experiment we have measured the in-situ structural evolution of the SrCo_{1-x}M_xO_{3-δ} system (M = V, Nb and Ti, x= 0.05) in the usual working conditions of a cathode in a SOFC (in air from 200 °C to 800 °C). Neutron powder diffraction (NPD) data were collected in the diffractometer D2B. The high intensity mode ($\Delta d/d \approx 5 \cdot 10^{-4}$) was selected, with a neutron wavelength $\lambda = 1.594 \text{ \AA}$ within the angular 2θ range from 8° to 155°. About 2 g of the sample was contained in a quartz tube open to the ambient atmosphere, placed in the isothermal zone of a furnace with a vanadium resistor operating under vacuum ($P_{O_2} \approx 10^{-6}$ Torr). The measurements were carried out in air at 200, 400, 600 and 800°C. The collection time was of 3 h per pattern. The irregular background coming from the quartz container was interpolated from points devoid of reflections.

Fig. 1 illustrates the example of SrCo_{0.95}Ti_{0.05}O_{3-δ} at 400 °C and 800 °C. The crystal structure can be Rietveld-refined in the tetragonal *P4/mmm* space group till 400 °C (Fig. 2). At this temperature a tetragonal (*P4/mmm*) to cubic (*Pm-3m*) phase transition takes place. Fig. 2 shows the crystal structures of the tetragonal and cubic phases. The thermal evolution of the lattice parameters and the oxygen content can be seen in Fig. 3. The tetragonal superstructure is due to the long-distance ordering of the vacancies along the c-axis. The occupancy factors of O atoms strongly decrease (Fig. 3b) as temperature

increases, which favours the oxygen ion mobility, as required in SOFC cathodes. Similar behaviours have been observed in the remaining materials.

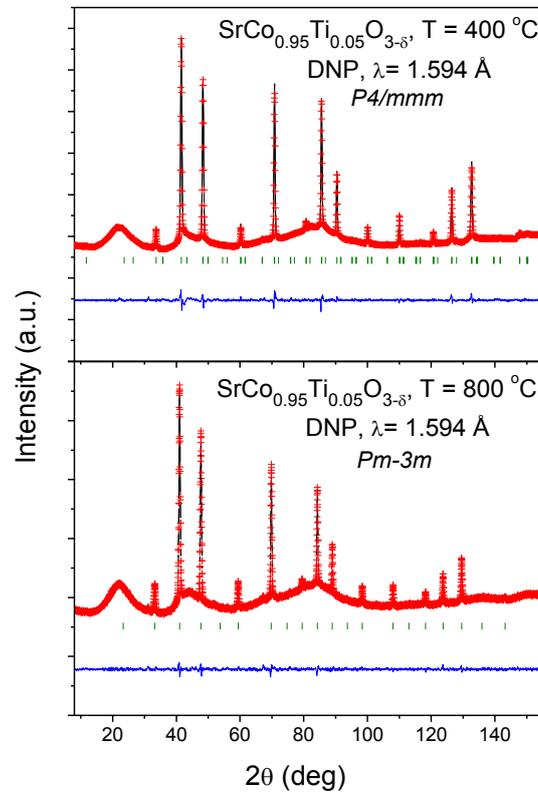


Fig. 1. Observed (crosses), calculated (full line) and difference (at the bottom) NPD profiles for $\text{SrCo}_{0.95}\text{Ti}_{0.05}\text{O}_{3-\delta}$ at $400\text{ }^\circ\text{C}$ and $800\text{ }^\circ\text{C}$. The vertical markers correspond to the allowed Bragg reflections.

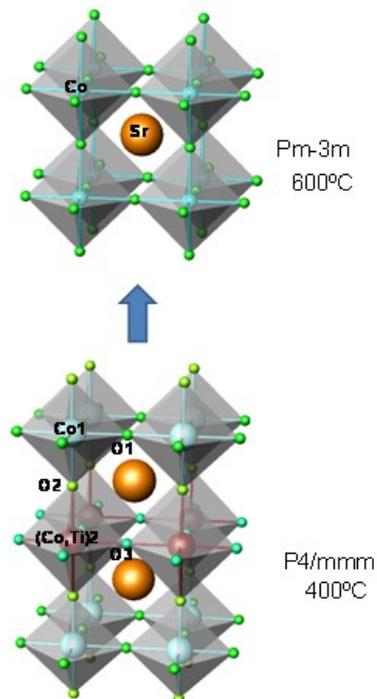


Fig. 2. Crystal structures of the tetragonal and cubic phases

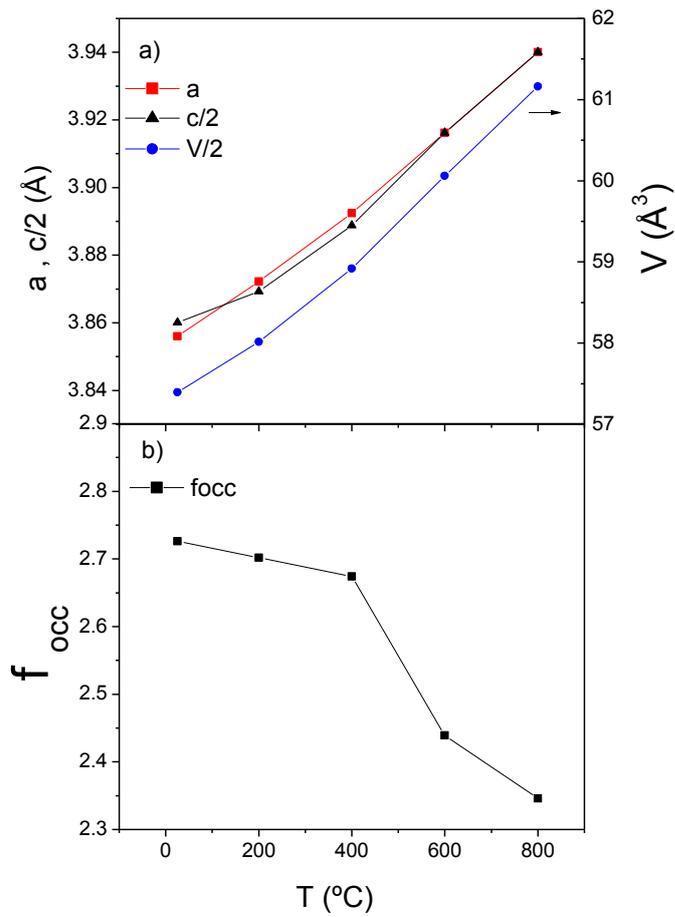


Fig. 3 Thermal evolution of structural parameters and oxygen content.

References

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- [2] Z. Q. Deng, W. Liu, C. S. Chen, H. Lu, W. S. Yang, *Solid State Ionics* 170 (2004) 187.
- [3] P. Zeng, R. Ranj, Z. Chen, W. Zhou, H. Gu, Z. Shao, S. Liu, *J. Alloys and Comp.*, 455 (2008) 465-470.
- [4] V. Cascos, L. Troncoso, J.A. Alonso, *International Journal of Hydrogen Energy*, 40 (2015) 11333-11341