

Experimental report

23/09/2024

Proposal: 5-24-720

Council: 4/2023

Title: In situ investigations of the hydrogenation of Ni₃In

Research area: Chemistry

This proposal is a new proposal

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Samples: Ni₃Ga
Ni₃In

Instrument	Requested days	Allocated days	From	To
D20	3	3	24/11/2023	27/11/2023

Abstract:

In situ neutron powder diffraction will reveal the hydrogen (deuterium) positions and occupations in the hydrides Ni₃InH_x(D)_x (hexagonal filled Mg₃Cd type) and the intermediate cubic Ni₃InH_z (filled AuCu₃ type). The diffraction experiment will be performed in a user-supplied sapphire single crystal cell at deuterium pressures up to 70 bar and temperatures up to 750 K. Rietveld analysis of diffraction data will allow the location of deuterium positions in the deuterides and the mapping of the complete reaction pathway of deuterium uptake and release.

***In situ* investigations of the hydrogenation of Ni₃In (5-24-720)**

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A variety of nickel-rich intermetallic compounds MNi_3 compounds are reported in the literature, e. g. for $M = \text{Mg, Al, Ga, Ge, Mn, Fe}$ (AuCu₃ type), Zr, In, Sn (Ni₃Sn type), Hf (BaPb₃ type), V, Nb, Ta (TiAl₃ type), Nb, Mo, Sb (Cu₃Ti type), Ti (TiNi₃ type), lanthanides Ln (CeNi₃ type, PuNi₃ type). Except for nickel-rich lanthanide compounds, which are well-studied for their ability to form hydrides in distorted variants of the hydrogen-free parent intermetallic structures [1], nickel-containing hydrides with p-block or transition metals are less well known.

Ni₃In (Ni₃Sn structure type) was studied *in situ* by neutron powder diffraction data under deuterium pressure (D20 in high-resolution mode at $\lambda = 1.87 \text{ \AA}$, 2 min per pattern, sapphire single-crystal cell [2–5]) according to the following temperature-pressure protocol (**Figure 1**):

- 1) 7.7 MPa D₂: 300 K $\rightarrow$ 680 K $\rightarrow$ 300 K
- 2) 2.7 MPa D₂: 300 K $\rightarrow$ 700 K $\rightarrow$ 300 K
- 3) 0.9 MPa D₂: 300 K $\rightarrow$ 800 K $\rightarrow$ 300 K

The deuteration reaction of Ni₃In starts at around 395 K with the formation of $\alpha\text{-Ni}_3\text{InD}_x$, leading to a significant increase in the c lattice parameter and a unit cell volume expansion of around 2.1%. Possible deuterium sites are [Ni₆] octahedral voids. The deuterium positions were established by alternating refinement of deuterium positions or occupation, followed by Fourier analysis. The $\alpha\text{-Ni}_3\text{InD}_x$ crystallizing in the hexagonal-filled Ni₃Sn structure type and deuterium was found to occupy the [Ni₆] octahedral voids (**Figure 2, Table 1**).

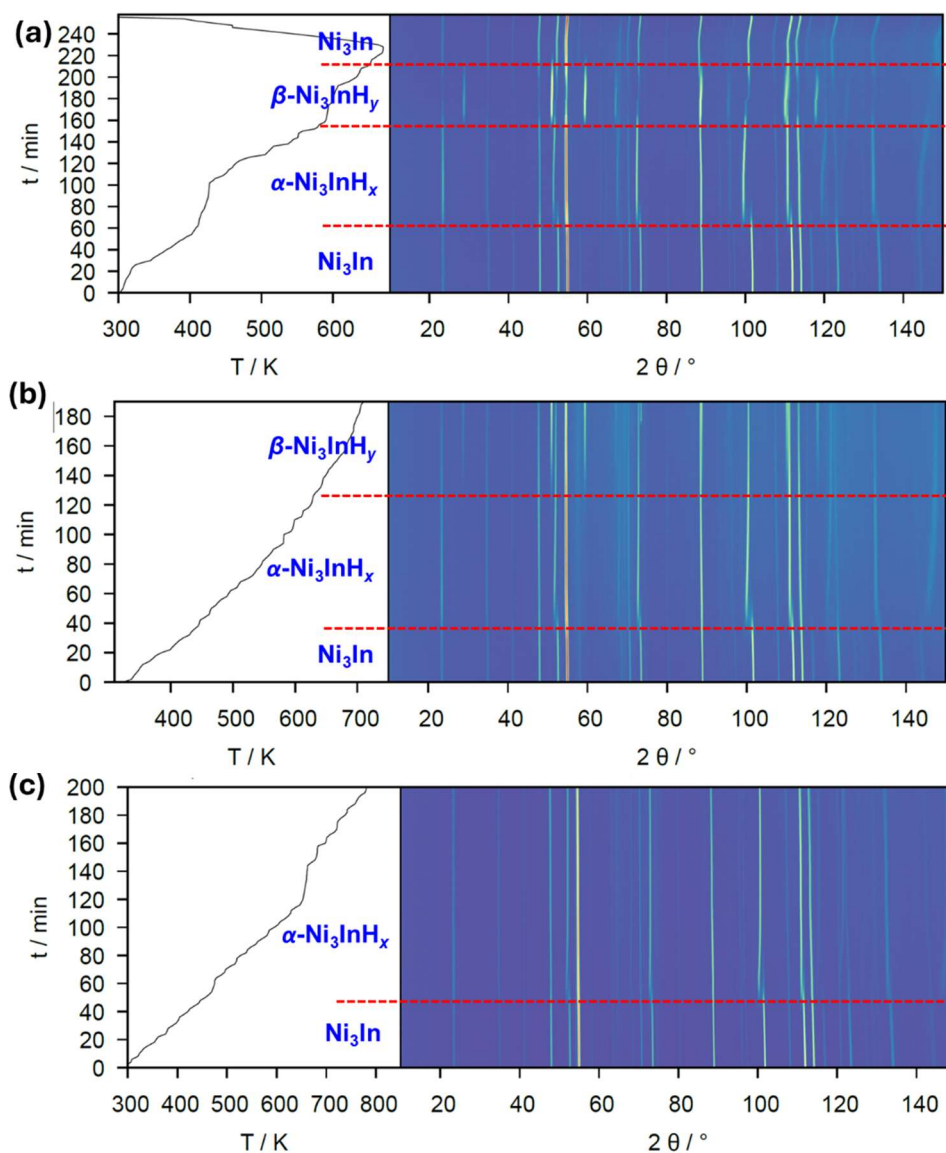


Figure 1: *In situ* neutron powder diffraction data of the deuteration of Ni₃In taken on diffractometer D20 at $\lambda = 1.86897(6) \text{ \AA}$ in a single crystal sapphire cell under various deuterium pressures and temperature conditions: (a) 7.7 MPa D₂; (b) 2.3 MPa D₂; (c) 0.9 MPa D₂. Intensities are in false colors.

Table 1: Refined atomic parameters of α -Ni₃InD_{0.65(3)} ($P6_3/mmc$, $a = 5.3427(9)$ Å, $c = 4.3332(8)$ Å).

atom	Wyck.	x	y	z	$B / 10^4 \text{ pm}^2$	Occ.
In1	2d	1/3	2/3	3/4	1.534(6)	1
Ni1	6h	0.1624(2)	0.3249(2)	1/4	1.424(8)	1
D	2a	0	0	0	2.589(3)	0.65(3)

Table 2: Refined atomic parameters of β -Ni₃InD_{0.81(5)} ($Pm\bar{3}m = 3.7838(5)$ Å).

atom	Wyck.	x	y	z	$B / 10^4 \text{ pm}^2$	Occ.
In1	1a	0	0	0	1.818(8)	1
Ni1	3c	1/2	1/2	0	1.454(4)	1
D	1b	1/2	1/2	1/2	2.485(3)	0.81(5)

Throughout heating the development of the α -Ni₃InD_x corresponds to a reduction of the Ni₃In phase fraction until around 425 K, where we exclusively observe the formation of the α -Ni₃InD_{0.65(3)}. The phase fraction of the hexagonal α -Ni₃InD_x phase remains constant until 550 K, when a new deuterated phase, β -Ni₃InD_y forms (filled AuCu₃ type, $Pm\bar{3}m$ $a = 3.78431(4)$ Å, **Figure 3, Table 2**).

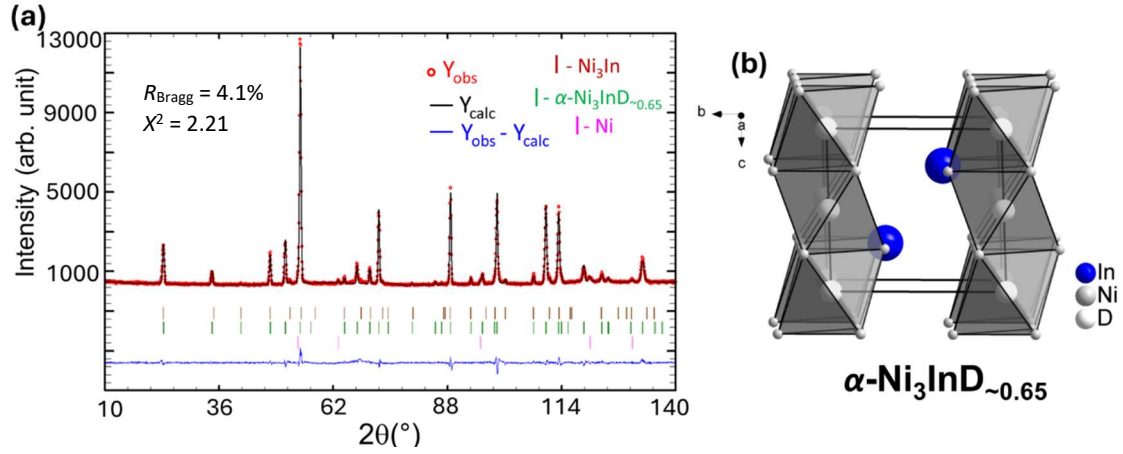


Figure 2: (a) Rietveld refinement and (b) crystal structure of α -Ni₃InD_{0.65(3)} (D20, ILL, Grenoble, $\lambda_{\text{neutrons}} = 1.8689(2)$ Å, 7.7 MPa deuterium gas pressure, $T = 422$ K)

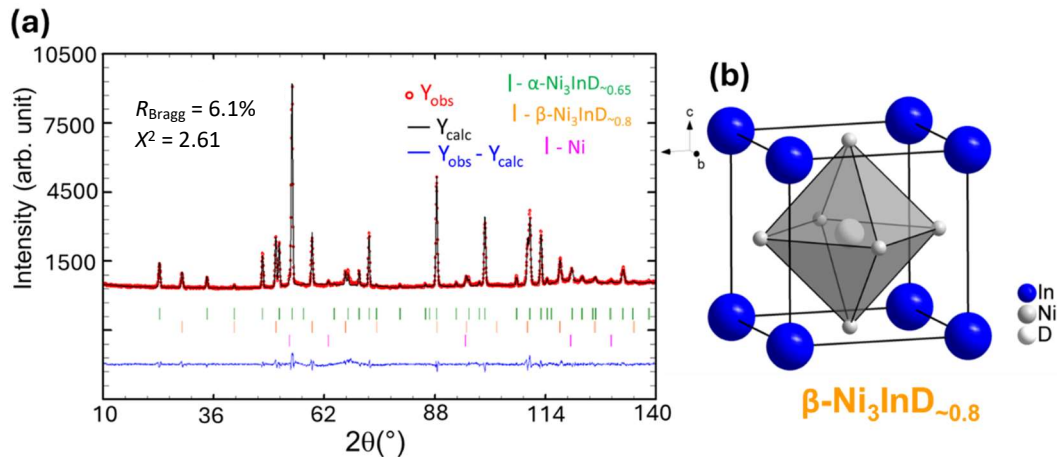


Figure 3: (a) Rietveld refinement and (b) crystal structure of β -Ni₃InD_{0.8(5)} (D20, ILL, Grenoble, $\lambda_{\text{neutrons}} = 1.8689(2)$ Å, 7.7 MPa deuterium gas pressure, $T = 581$ K)

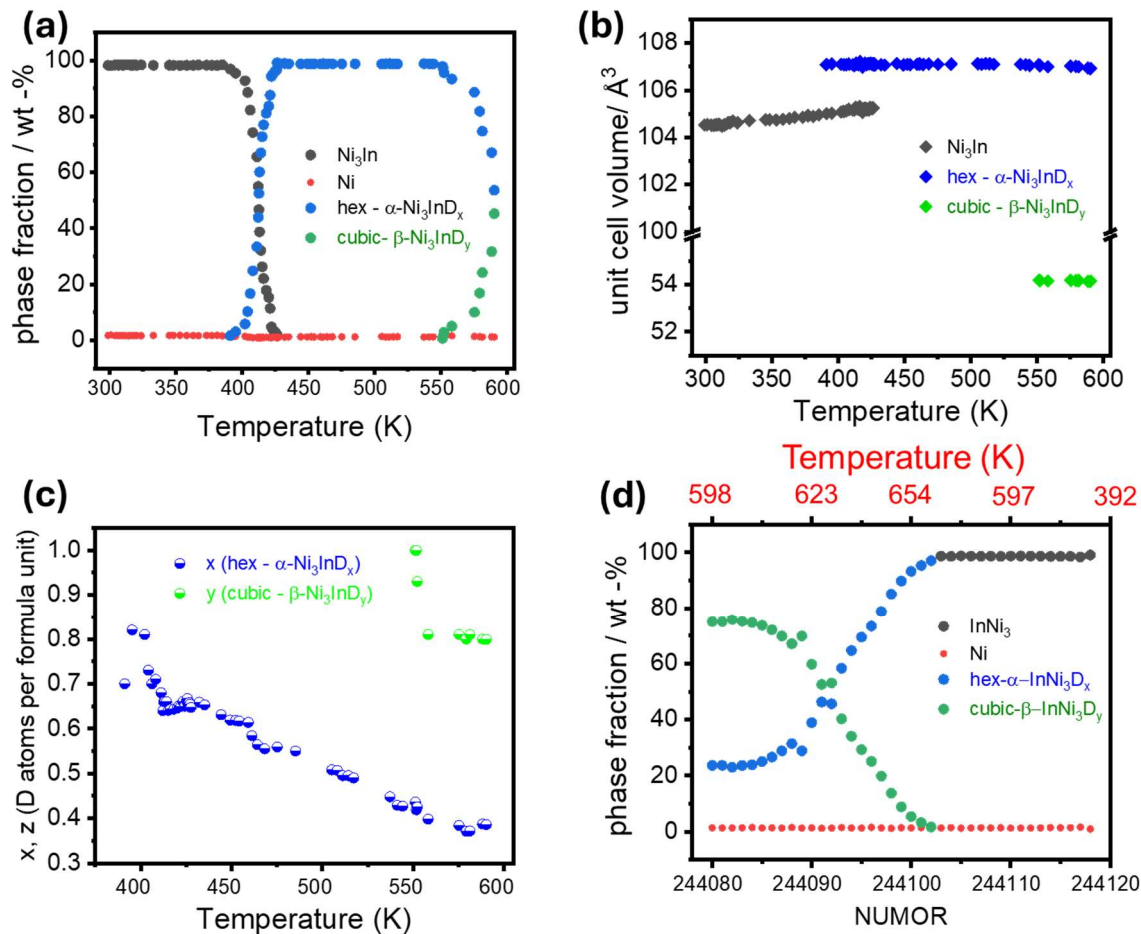


Figure 4: Control parameters of the *in situ* deuteration of Ni_3In and refined structural parameters based on neutron powder diffraction data measured at D20 ($\lambda = 1.86897(6)$ Å) at 7.7 MPa D_2 and various temperatures: **(a)** phase fractions – upon heating, **(b)** unit cell volume – upon heating, **(c)** deuterium content per formula unit – upon heating, **(d)** phase fractions – upon cooling under vacuum. Each NUMOR 2 min.

At $550 \text{ K} \leq T \leq 590 \text{ K}$, a miscibility gap between α - and β -phase is presumed, as both phases are observed simultaneously, and $\alpha\text{-Ni}_3\text{InD}_x$ has a sharp decrease of ≈ 0.44 deuterium atoms per formula unit. At around 590 K, we detected a reduction in deuterium pressure, suggesting a potential leak in the cell. Therefore, we decided to apply vacuum, heat the cell to 670 K and cool it down to room temperature. Upon cooling under vacuum, both hydrides decomposed around 665 K and the parent compound hexagonal Ni_3In formed (**Figure 4**).

References:

- [1] Y. E. Filinchuk, K. Yvon, *J. Alloys Compd.* **2005**, 404-406, 89-94.
- [2] R. Finger, N. Kurtzemann, T. C. Hansen, H. Kohlmann, Design and use of a sapphire single-crystal gas-pressure cell for *in situ* neutron powder diffraction, *J. Appl. Crystallogr.* **2021**, 54, 839–846.
- [3] A. Götze, J. Möllmer, H. Kohlmann, *Inorg. Chem.* **2018**, 57, 10925–10934.
- [4] A. Götze, H. Auer, R. Finger, T. C. Hansen, H. Kohlmann, A sapphire single-crystal cell for *in situ* neutron powder diffraction of solid-gas reactions, *Phys. B (Amsterdam, Neth.)* **2018**, 551, 395–400.
- [5] A. Götze, S. C. Stevenson, T. C. Hansen, H. Kohlmann, *Crystals* **2022**, 12, 1704.