Experimental report

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Fe2VAl could become a good intermotecture material if its thermal conductivity could be reduced. This can be achieved by synthesizing Fe2VAl under nanograined (~ 100 nm) form. But this leads to a decrease of its Seebeck coefficient and thus, to poor thermoelectric properties. To understand this opposite effect, we undertook a neutron diffraction experiment (9/9/2020, CRG-2769) on D1B to determine as a function of temperature (25 – 900°C) the crystalline structures adopted by nanograined-Fe2VAl after ball-milling: disordered structures could indeed lead to low Seebeck values. A rapid examination of our data shows that at 25°C, ball-milled Fe2VAl crystallizes in the disordered A2 structure. Upon heating, it orders successively in the B2 and L21 (fully ordered) structures around 300° C - 400°C respectively. Unfortunately, an uncontrollable ramp of the high- temp. furnace prevented us from accurately measuring the temperature of these irreversible transitions. In order to finish this study with publishable data, we hence require an extra beam time (12 – 24h) on D1B equipped with the better controllable 25 - 500°C furnace to measure a new sample prepared under the same conditions.

Easy Access 724, 1/2/2021 : Crystal structures in nanograined-Fe₂VAI polycrystals for thermoelectric applications

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Fe₂VAl could become a good thermoelectric material if its thermal conductivity could be reduced. This can be achieved by synthesizing Fe₂VAl under meso-size grains (~ 1 µm) and well densified polycrystals. But this leads to a decrease of its Seebeck coefficient and thus, to poor thermoelectric properties. To understand this opposite effect, we undertook a neutron diffraction experiment (9/9/2020, CRG-2769) on D1B to determine as a function of temperature (25 – 900°C) the crystalline structures adopted by nanograined-Fe₂VAl after ball-milling: disordered structures could indeed lead to low Seebeck values. A rapid examination of our data shows that at 25°C, ball milled Fe₂VAl crystallizes in the disordered and metastable A2 structure. Upon heating, it orders successively in the B2 and L2₁ (fully ordered) structures around 300°C - 400°C respectively. Unfortunately, an uncontrollable ramp of the high- temp. furnace prevented us from accurately measuring the temperature of these irreversible transitions.

The in-situ experiment on a ball milled sample was carried out in the ILL medium-temperature furnace (25 - 550 °C). A 1 K / min ramp was programmed, and this allowed this time to follow precisely the A2 -> B2 and B2 -> L2₁ transitions. A plot of the normalised intensity of the 111, 200 and 220 lines is shown in Fig. 1. The 200 line is characteristic of the B2 partial order and the 111 line is characteristic of the L2₁ ordered structure. The A2 -> B2 transition occurs at 350 °C whereas the B2 -> L2₁ occurs at 550°C.

This implies that sintering the sample at 900 °C fully restores the L2₁ structure. As already stated in the CRG N° 2769 report, we finally related the poor Seebeck coefficient of ball milled and sintered samples to a contamination, which modifies the composition.

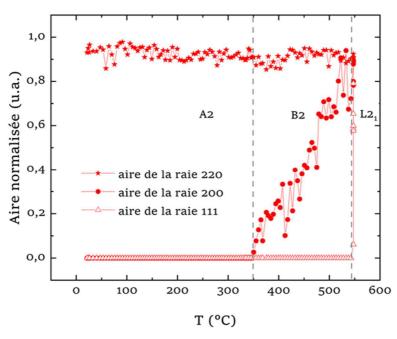


Fig. 1. Normalised integrated intensity of the 111, 200 and 220 lines of ball milled Fe₂VAI versus temperature.